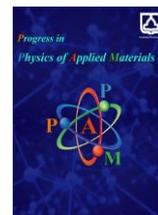




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## Synthesis and Investigation of Different Properties of $K_2FeO_4/ZnO$ and Its GO-Based Nanocomposites

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### ABSTRACT

This paper presents significant findings regarding the impact of ZnO and graphene oxide (GO) doping on the structural and electrical properties of potassium ferrate nanostructures. The samples were synthesized using a thermal treatment method at temperatures of 773, 873, and 973 K. The structural characteristics, optical and magnetic properties of the synthesized samples were analyzed using various techniques. The photocatalytic activity of  $K_2FeO_4/ZnO$  nanoparticles under visible light irradiation was investigated using methylene blue (MB) degradation as a probe reaction. The results indicated a significant enhancement in photocatalytic activity when GO was incorporated into the  $K_2FeO_4/ZnO$  nanocomposite. The dielectric constant and dielectric loss were measured at room temperature across a frequency range of 4 to 8 MHz using an LCR meter. Our findings reveal that the addition of graphene oxide (GO) did not result in a significant enhancement in dielectric permittivity compared to potassium ferrate/ZnO nanocomposites. Therefore, potassium ferrate-based nanocomposites, given their favorable dielectric properties, are promising candidates for a wide range of applications, including devices operating at microwave frequencies, various optical and microelectronic applications, and as materials for microwave absorption.

## 1. Introduction

The energy transition must address the dual challenge of meeting the demands of a growing, increasingly energy-intensive global economy while adhering to environmental and ecological constraints. In this context, there is a significant interest in the development of innovative environmental technologies for energy storage and recovery [1]. Composites have garnered considerable attention in recent years due to their potential to offer novel technological solutions with enhanced properties, particularly for advanced applications such as electronics, energy systems, and automotive technologies. For electrical applications specifically, the incorporation of metallic or ceramic micro/nanoparticles into a polymer matrix has

been shown to confer desirable properties [2]. With the continued trend toward greater integration, miniaturization, and multifunctionality in modern electronics, nanocomposite materials with high dielectric permittivity are expected to see broader use in capacitors, field-effect transistors, memory devices, and energy storage systems [1]. Among energy storage devices, ceramic-based dielectric capacitors are particularly valued due to their high temperature stability, superior mechanical properties, and their ability to store large amounts of energy with fast charge-discharge rates [3]. However, the electronic industry still heavily relies on lead-based dielectric materials, which contribute to environmental pollution. Additionally, the non-stoichiometric composition of these dielectrics often requires high processing temperatures. Lead-free materials

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such as CuO, ZnO, and TiO<sub>2</sub>, along with co-doping elements such as Fe, Na, Mg, Ag, Mn, and Al, have emerged as effective alternatives in this regard [4]. In fact, the chemical valence states of complex ions can be flexibly adjusted by regulating the ratio of co-doping ions. As the valence states of these complex ions change, it is expected that substitution types, relative concentrations of charge carriers, and the resulting dielectric behavior will also be altered. From the perspective of defect design, substituting complex ions with nonequivalent valence states can result in optimized dielectric properties, provided that overall charge balance is maintained [5]. The primary reason for introducing substitutional impurity ions is to increase the electrical resistivity of the composite, which is crucial for reducing electrical or dielectric losses, especially in high-frequency applications [6]. Among the various nanomaterials used in composite fabrication to enhance physical properties, graphene and graphene oxide (GO) are of particular importance due to their unique properties, such as exceptional stiffness, strength, high specific surface area, excellent thermal conductivity, and gas impermeability [7]. Graphene also exhibits high electron mobility and excellent electrical and chemical properties, which support the formation of nanoparticles in composite structures. Moreover, graphene's high dielectric properties contribute to low impedance and limited absorption characteristics, although its tendency to agglomerate can pose challenges [8]. In addition to their use in electronics and energy storage, nanocomposites, particularly those incorporating metal oxides like ZnO and graphene-based materials, are increasingly being investigated for their potential applications in various fields [9].

These materials can be employed in the development of smart fertilizers, which offer controlled release of nutrients and improve plant uptake efficiency. Zinc oxide (ZnO) nanoparticles, for instance, have been shown to enhance plant growth and yield by providing essential zinc nutrition more effectively than conventional fertilizers. Furthermore, graphene-based materials are being explored for their role in promoting seed germination, improving water retention in soil, and acting as carriers for agrochemicals, thereby reducing environmental pollution from excess chemical use [10]. Among the latest technologies, photocatalysis, an eco-friendly nanotechnology approach, offers a potentially sustainable solution to current agricultural environmental challenges. [11]. Therefore, researchers, have extensively investigated the photocatalytic degradation of water pollutants using various semiconductor oxide photocatalysts, including ZnO and TiO<sub>2</sub> [12, 13].

The simultaneous study of dielectric and photocatalytic properties in nanoparticles holds tremendous potential for advancing technologies in areas like water purification, solar energy conversion, and environmental remediation. Further investigation is crucial to unlock the full potential of these multifunctional nanomaterials. Accordingly, in this study, magnetic nanostructures of potassium ferrate -doped zinc oxide nanoparticles were synthesized using the thermal treatment method. In the next step, these nanoparticles were successfully anchored on graphene oxide sheets by hydrothermal method.

## 2. Experimental

### 2.1. Materials and Methodology

In the synthesis of K<sub>2</sub>FeO<sub>4</sub>/ZnO and K<sub>2</sub>FeO<sub>4</sub>/ZnO/GO deionized water as the solvent and metal nitrates as precursors and Polyvinylpyrrolidone (PVP) as the capping agent have been used. Iron nitrate, Fe (NO<sub>3</sub>)<sub>3</sub>•9H<sub>2</sub>O, and potassium nitrate, KNO<sub>3</sub>, Zinc nitrate, Zn(NO<sub>3</sub>)<sub>2</sub>, were purchased from Acros Organics (99%). The following reagents, all obtained from Merck, were used as received without further purification: polyvinyl alcohol (PVA, MW = 31,000 g/mol), hydrogen peroxide (30% H<sub>2</sub>O<sub>2</sub>), graphite, ethanol, hydrochloric acid (37% HCl), potassium permanganate (99.5% KMnO<sub>4</sub>), and sulfuric acid (98% H<sub>2</sub>SO<sub>4</sub>).

### 2.2. Synthesis of K<sub>2</sub>FeO<sub>4</sub>

The synthesis of potassium ferrate nanoparticles was achieved via a thermal treatment method. A solution of 4 g PVP in 100 mL of deionized water was prepared at 353 K. Subsequently, a solution containing 0.2 mmol potassium nitrate and 0.1 mmol iron nitrate (molar ratio K:Fe = 2:1) was added, and the mixture was stirred magnetically for 2 hours. Then, the mixture was dried for 24 h at 353K in an oven. The calcination of the powder was performed at 723, 823, and 923K for 3h to crystallize nanocrystal and decompose organic material.

### 2.3. Synthesis of K<sub>2</sub>FeO<sub>4</sub>/ZnO

To synthesize K<sub>2</sub>FeO<sub>4</sub>/ZnO nanoparticles, again thermal treatment method was used. In this way that, 0.1 g of synthesized K<sub>2</sub>FeO<sub>4</sub> nanoferrate was dispersed into 500 ml deionized water. The solution was placed in an ultrasonic unit for 30 min and then, 0.4 g of zinc nitrate and 19 g of PVP were slowly added to it and the mixture was dried 24 h at 353K in an oven. Finally, the powder was calcined at 723, 823, and 923K for 3h.

### 2.4. Synthesis of Graphene Oxide

Graphene oxide (GO) was synthesized via a modified Hummers method [14]. 1 g of graphite powder was added to a mixture of 3 g potassium permanganate (KMnO<sub>4</sub>, 99.5% purity) and 25 mL concentrated sulfuric acid (H<sub>2</sub>SO<sub>4</sub>, 98% purity). This mixture was stirred for 3 hours at 20 °C. The resulting suspension was then diluted by slow addition of 100 mL deionized water and 5 mL of 30% hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>). After 5 minutes of stirring, a bright yellow solution formed. This solution was centrifuged at 5000 rpm for 15 minutes, and the obtained solid was washed several times with 10% hydrochloric acid (HCl) and distilled water. The resulting yellow-brown precipitate was collected via filtration and dried under vacuum at 333 K for 24 hours.

### 2.5. Synthesis of K<sub>2</sub>FeO<sub>4</sub>/ZnO /GO

K<sub>2</sub>FeO<sub>4</sub>/ZnO/GO nanocomposites were prepared as follows: A dispersion of 2 g of GO in a 1:1 mixture of distilled water and ethanol (100 mL total volume) was prepared. Separately, 1.5 g of K<sub>2</sub>FeO<sub>4</sub>/ZnO nanopowder was dispersed in 100 mL of distilled water. The K<sub>2</sub>FeO<sub>4</sub>/ZnO dispersion was then slowly added to the GO

dispersion, and the resulting mixture was sonicated for 30 minutes. After sonication, the mixture was stirred for 24 hours. The resulting suspension was then centrifuged to separate the solid product, washed repeatedly with distilled water, and finally dried under vacuum at 60 °C for 24 hours to yield the  $K_2FeO_4/ZnO/GO$  nanocomposite.

### 3. Characterization Techniques

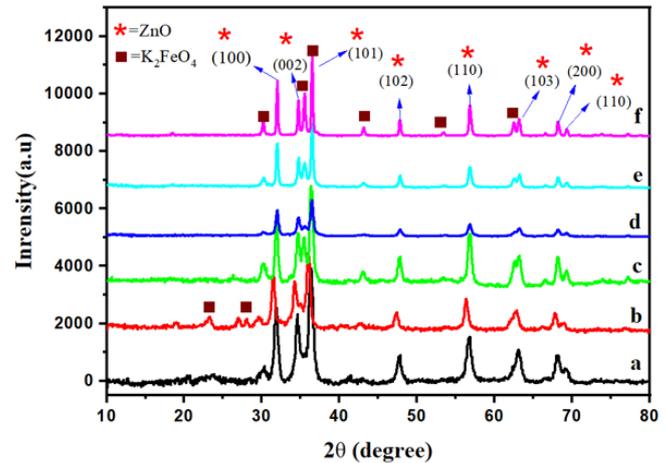
Morphological and structural properties of nanoferrate composites were investigated by X-ray diffraction (XRD, Panalytical X'Pert Pro-model from Netherlands,  $\lambda=1.54 \text{ \AA}$ ), field emission-scanning electron microscopy (FESEM, Zeiss SIGMA VP from Germany) images TE-SCAN type instrument coupled with Energy dispersive X-ray spectroscopy (EDX) for elemental analysis, transmission electron microscopy (TEM, Zeiss EM10C; accelerating voltage 100 kV, Germany), and Fourier transform infrared (FT-IR, PerkinElmer Spectrum Two, USA) spectroscopy. The Digimizer image analysis software was used for calculating the average particle sizes. The band gap energies of the  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites were analyzed by UV-visible absorption spectroscopy (UV1650PCSHIMADZU). The magnetic properties of all the synthesized samples were evaluated at room temperature via vibrating sample magnetometer (VSM) (VSM, DKB model), the magnetic characteristics were measured, and by using a laser Raman spectrophotometer (UniRAM, wavelength of excitation laser: 785 nm), Raman spectra were measured. Dielectric and impedance spectroscopy measurements were carried out in frequency range of 4Hz to 8 MHz using LCR meter (Model: HIOKI IM3536).

## 4. Results and Discussion

### 4.1. XRD Analysis

The crystal structure and phase composition of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites were examined using X-ray diffraction (XRD), as presented in Figure 1. In the XRD patterns of  $K_2FeO_4/ZnO$  nanocomposites, measured at three different temperatures (Figs. 1a, b, and c), peaks marked with squares correspond to the characteristic crystalline planes of  $K_2FeO_4$ , aligning with the orthorhombic structure (ICCD: 00-025-0652) [15]. Additional peaks observed in the XRD patterns correlate with the (100), (002), (101), (102), (110), (103), (200), and (112) crystallographic planes, confirming the presence of ZnO in the hexagonal wurtzite phase (JCPDS No. 36-1451) [16]. For the  $K_2FeO_4/ZnO/GO$  nanocomposite (Figs. 1d, e, and f), similar diffraction patterns were observed, suggesting the preservation of  $K_2FeO_4$  and ZnO phases, while the absence of the GO peak indicates complete exfoliation of GO, likely due to the integration of  $K_2FeO_4/ZnO$  particles within the GO matrix [17]. The average crystallite sizes for both  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites were calculated using the Scherrer equation  $D=K\lambda/\beta\cos\theta$ , where D represents the average crystallite size, K is the shape factor (0.9),  $\lambda$  is the wavelength of the X-ray radiation,  $\beta$  is the full width at half

maximum (FWHM) of the peak, and  $\theta$  is the diffraction angle [18]. The calculated crystallite sizes are presented in Table 1.



**Fig. 1.** XRD patterns of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.

### 4.2. FT-IR Analysis

Fourier-transform infrared (FTIR) spectroscopy was conducted to elucidate the structural properties, identify functional groups, and confirm the attachment of  $K_2FeO_4/ZnO$  nanoparticles onto GO sheets, as illustrated in Figure 2. Peaks observed at approximately  $471 \text{ cm}^{-1}$  and  $544 \text{ cm}^{-1}$  in Figure 2a, b, c, d, e, and f correspond to characteristic vibrations of the Fe–O bond [19]. Additionally, metal-oxygen (M–O) bonds (where M represents Fe, K, or Zn) are indicated by peaks around  $598 \text{ cm}^{-1}$ , supporting the successful formation of the  $K_2FeO_4/ZnO$  nanocomposite [20]. Absorption bands within the  $1400\text{--}2860 \text{ cm}^{-1}$  range, found in all samples, likely indicate the presence of residual PVP surfactant from the synthesis process [21].

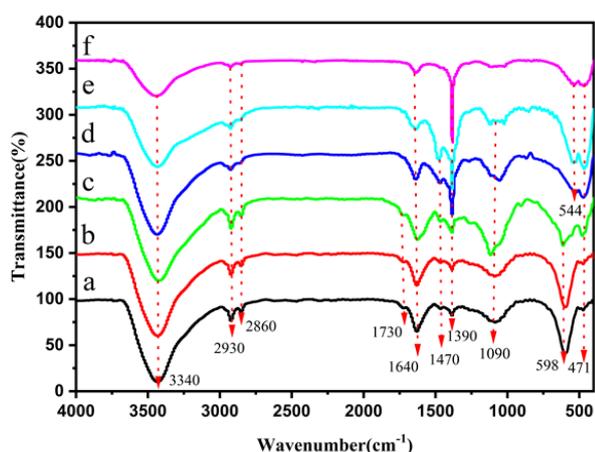
A strong band at  $3340 \text{ cm}^{-1}$  is attributed to O–H stretching, suggesting the presence of OH and/or COOH functional groups within the composite structure [22].

The peak at  $1090 \text{ cm}^{-1}$  corresponds to the ketone C–C–O stretch within the aryl ether ring, while the phenol O–H bending appears at  $1390 \text{ cm}^{-1}$  [23].

Additionally, peaks at  $1640 \text{ cm}^{-1}$  and  $1730 \text{ cm}^{-1}$  are associated with the aromatic C=C skeletal vibration and C=O stretching, respectively [24].

Furthermore, although some absorption bands related to residual PVP and GO functional groups (e.g., C=O, C–O, and O–H) are still detectable, the appearance of distinct metal–oxygen bands (Fe–O around  $471$  and  $544 \text{ cm}^{-1}$ , and Zn–O near  $598 \text{ cm}^{-1}$ ) clearly confirms the successful formation of the  $K_2FeO_4/ZnO$  composite structure and its effective anchoring onto the GO sheets.

These observations are consistent with previous reports and demonstrate the coexistence of precursor remnants and the final nanocomposite framework.



**Fig. 2.** FT-IR patterns of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.

#### 4.3. FE-SEM, EDX, and TEM Analyses

The microstructure and morphology of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites were examined using field-emission scanning electron microscopy (FE-SEM) and transmission electron microscopy (TEM), as shown in Figures 3, 4, and 5. FE-SEM images (Fig. 3) reveal that  $K_2FeO_4/ZnO$  nanoparticles exhibit a spherical-like morphology with varying degrees of particle aggregation. To investigate the elemental composition of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites, EDX spectroscopy and EDS mapping were performed. Figure 4(a and b), provides insight into the spatial distribution and elemental composition of Zn, Fe, O, K, and C. The EDX spectrum of the  $K_2FeO_4/ZnO$  nanocomposite revealed the presence of Zn, Fe, K and O with weight percentages of 51.78%, 41.17%, 5.71%, and 1.34%, respectively (Fig. 4a). These results confirm that the nanocomposite is primarily composed of those four elements. Similarly, the EDX analysis of the  $K_2FeO_4/ZnO/GO$  nanocomposite showed the presence of Zn, Fe, K, and C with weight percentages of 44.97%, 8.96%, 25.08%, and 20.89%, respectively (Fig. 4b), confirming the successful incorporation of graphene oxide.

**Table 1.** Average particle sizes of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  calcined at different temperatures determined by XRD, and FESEM techniques.

Samples	Average particle size XRD (nm)	Average particle size FESEM (nm)
$K_2FeO_4/ZnO$ (T=723 K)	19	28
$K_2FeO_4/ZnO$ (T=823 K)	26	34
$K_2FeO_4/ZnO$ (T=923 K)	32	41
$K_2FeO_4/ZnO/GO$ (T=723 K)	17	25
$K_2FeO_4/ZnO/GO$ (T=823 K)	22	29
$K_2FeO_4/ZnO/GO$ (T=923 K)	27	35

In contrast, the  $K_2FeO_4/ZnO/GO$  nanocomposite displays small  $K_2FeO_4/ZnO$  particles distributed on and within the GO sheet surfaces, which is particularly evident in the TEM images (Fig. 5b). The TEM analysis of  $K_2FeO_4/ZnO$  (Fig. 5a) shows near-spherical nanoparticles with some agglomeration. This aggregation is likely due to strong interparticle interactions, including van der Waals forces and magnetic dipolar interactions, which can enhance the magnetic coupling within the nanocomposite [21].

#### 4.4. UV-Vis Analysis

UV-Vis absorption spectra of the synthesized potassium ferrate nanocomposites doped with ZnO and GO were recorded over a wavelength range of 200–800 nm, as shown in Figure 6. The spectra display distinct absorption peaks attributed to electronic transitions within the material. For undoped potassium ferrate, the absorption

edge lies in the UV region, with a sharp increase in absorbance below 400 nm, corresponding to the material's band gap absorption from the valence to the conduction band. Upon incorporation of ZnO and GO, a noticeable red-shift in the absorption edge occurs, indicating a reduction in band gap energy. This red-shift suggests successful integration of ZnO and GO within the potassium ferrate matrix, resulting in electronic structure modifications and enhanced light absorption in the visible region. The broader absorption bands in the visible spectrum further imply the formation of sub-bandgap states introduced by the dopants, which may improve the photocatalytic and optoelectronic properties of the material. The band gap energy ( $E_g$ ) of the samples was calculated using the Tauc plot method, revealing a decrease in  $E_g$  with higher dopant concentrations. This reduction in band gap energy makes the nanocomposites suitable for applications in optoelectronics, visible-light photocatalysis, and energy storage devices. Changes in the energy gap can be better

understood if one considers that the interatomic spacing increases when the amplitude of the atomic vibrations increases due to the increased thermal energy [25]. On the other hand the observed red-shift in the absorption edge and the corresponding reduction in band gap energy with GO incorporation can be attributed to the electronic coupling and interfacial interactions between GO and the  $K_2FeO_4/ZnO$  nanostructure. GO, acting as an electron acceptor, introduces localized states within the band structure and promotes enhanced charge delocalization. Additionally, the increase in particle size at higher calcination temperatures (as confirmed by XRD and TEM data) contributes to band gap narrowing due to reduced quantum confinement. These effects collectively explain the observed trend in band gap reduction.

#### 4.5. VSM Analysis

Vibrating Sample Magnetometry (VSM) was used to investigate the magnetic properties of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites calcined at 773, 873, and 973 K. The magnetization curves (M-H loops) provide insights into the magnetic behavior of these materials under an applied magnetic field.

**Saturation Magnetization ( $M_s$ ):** The magnetization curves show varying saturation magnetization ( $M_s$ ) values across the samples. For  $K_2FeO_4/ZnO$ ,  $M_s$  values are 3.87,

3.75, and 2.93 emu/g for calcination temperatures of 773, 873, and 973 K, respectively. The  $K_2FeO_4/ZnO/GO$  samples show  $M_s$  values of 2.8, 2.76 and 1.65 emu/g for the same temperatures. The gradual decrease in  $M_s$  with increasing calcination temperature is attributed to temperature-induced microstructural changes. At higher calcination temperatures, although the overall crystallinity improves due to organic removal and grain growth, the formation of microstructural defects (e.g., oxygen vacancies or interfacial strains) may increase, which can locally affect magnetic ordering and reduce saturation magnetization. These subtle structural variations, although beneficial for charge transport and photocatalysis, can disrupt magnetic coherence and reduce the net magnetic response.

**Comparison of Samples:**  $K_2FeO_4/ZnO$  nanocomposites exhibit higher saturation magnetization compared to  $K_2FeO_4/ZnO/GO$  samples. This difference may stem from the presence of graphene oxide, which decreases the magnetic iron oxide content and introduces diamagnetic behavior, thereby diluting the overall magnetic response of the composite.

**Magnetic Hysteresis:** The narrow magnetic hysteresis loops observed indicate low coercivity and soft magnetic behavior in the nanocomposites, making them suitable for applications in magnetic memory devices and catalysis.

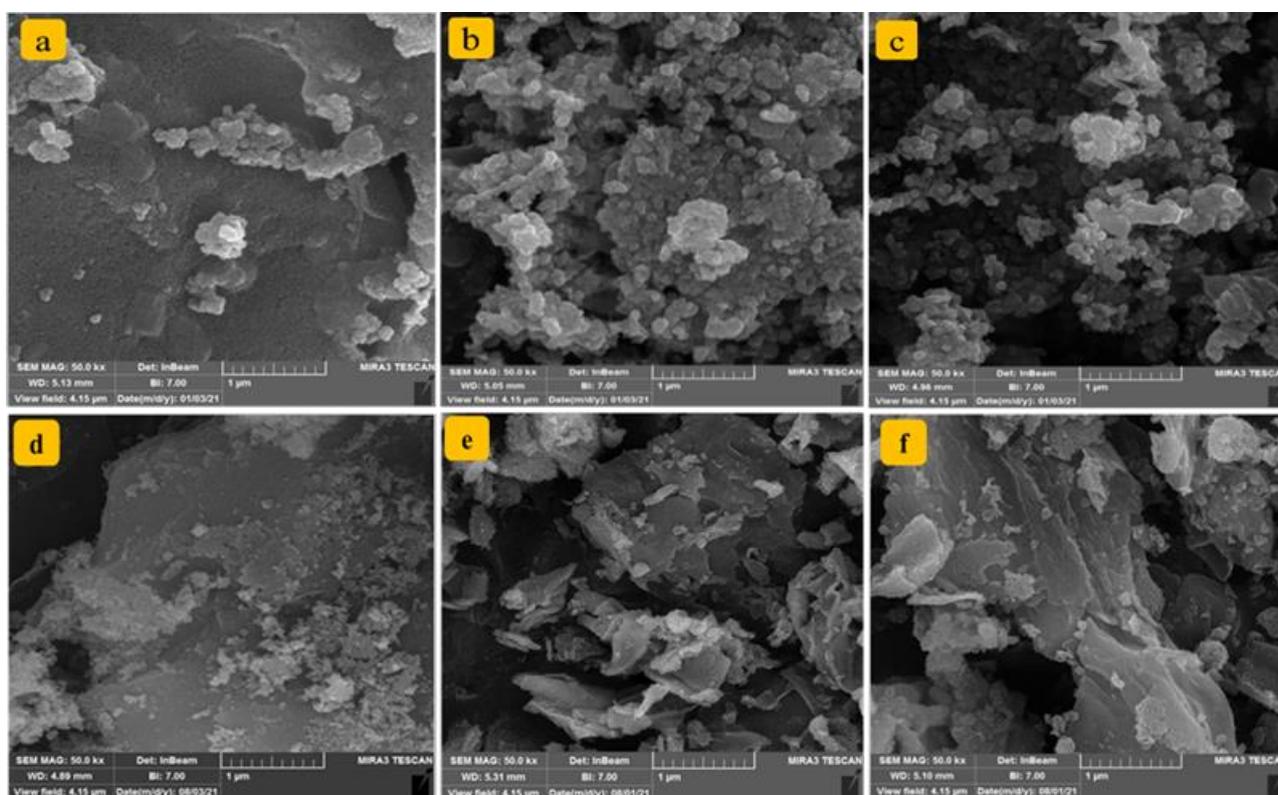


Fig. 3. FESEM images of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.

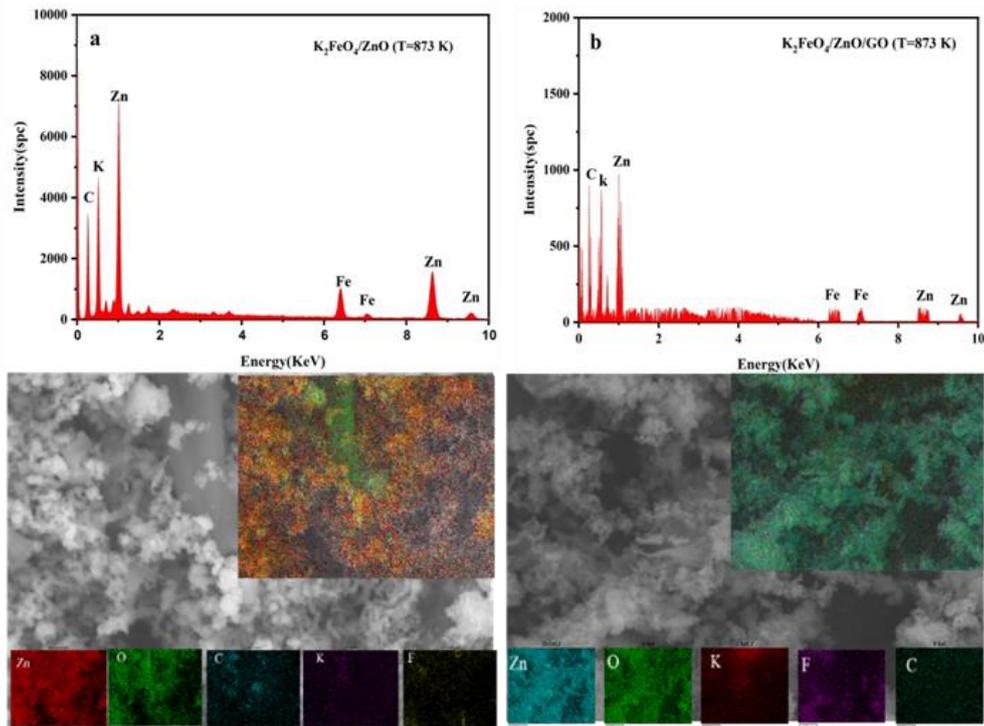


Fig. 4. EDX and map images of  $K_2FeO_4/ZnO$  ((a) T=823K), and  $K_2FeO_4/ZnO/GO$  ((b) T=823K).

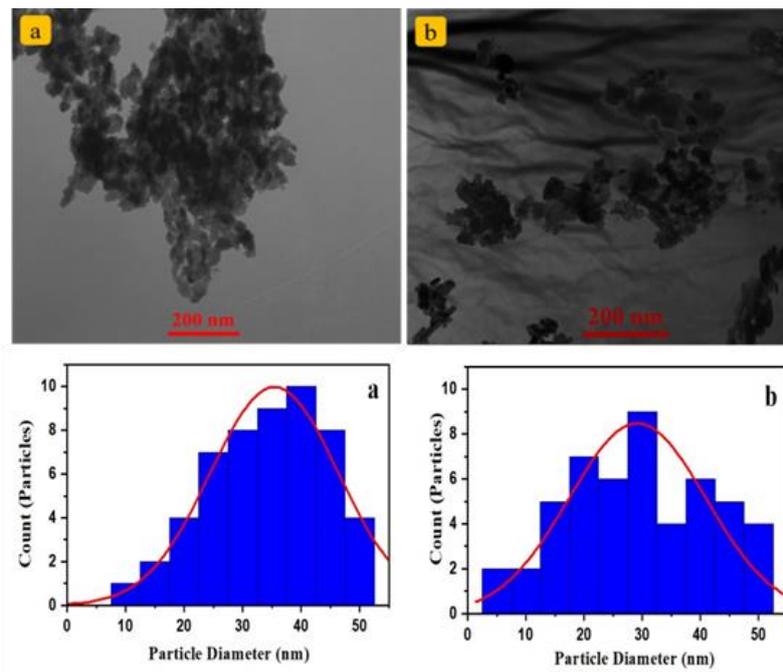


Fig. 5. TEM image and particle size distribution histograms (a)  $K_2FeO_4/ZnO$  calcined at 923 and (b)  $K_2FeO_4/ZnO/GO$  calcined at 923 K.

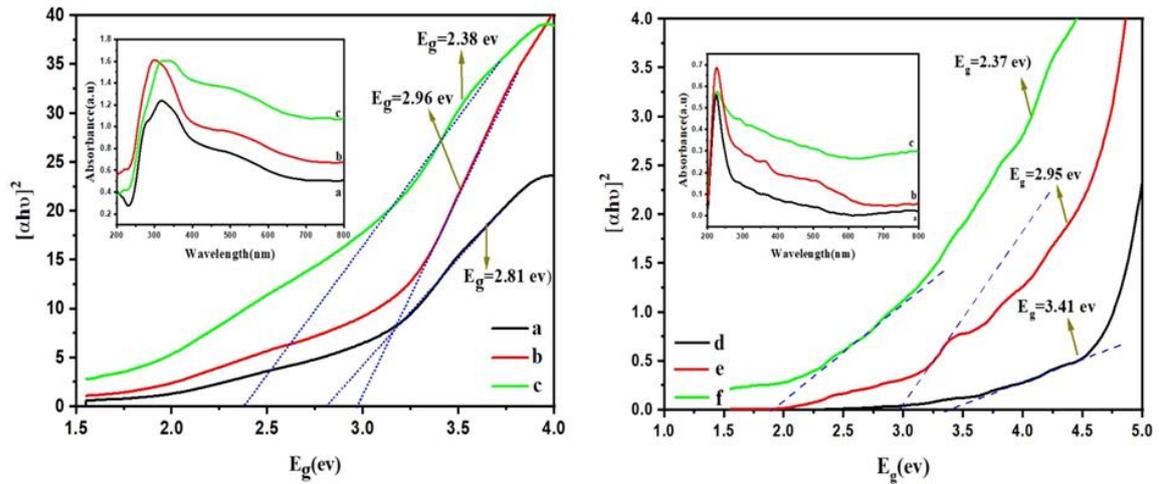


Fig. 6. UV patterns of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.

**Effect of Temperature:** As the calcination temperature increases, a decrease in saturation magnetization is observed, suggesting a decline in magnetic properties at higher temperatures. This effect may be attributed to reduced crystallite size and structural alterations induced by thermal treatment, which weaken magnetic interactions within the material [26]. VSM analysis reveals that both  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites possess notable magnetic properties influenced by temperature and composition (Fig. 7). The higher saturation magnetization at lower calcination temperatures suggests these materials maintain superior magnetic properties when processed at lower temperatures. While the inclusion of GO reduces overall magnetization, it provides additional functionalities beneficial for applications such as magnetic storage and catalysis. A summary of magnetic parameters obtained from VSM spectra is provided in Table 2.

#### 4.6. RAMAN Analysis

Raman spectroscopy was used to analyze the structural properties of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites calcined at temperatures of 773, 873, and 973 K, as shown in Figure 8. The spectra display characteristic peaks corresponding to ZnO, graphene oxide (GO), and iron oxide vibrational modes. Specifically, peaks at 370, 459, 644, and 840  $cm^{-1}$  are associated with the  $A_1(TO)$ ,  $E_1(TO)$ , and  $E_2(\text{high})$  vibrational modes of ZnO [27]. The D and G bands of GO, prominent around 1352 and 1617  $cm^{-1}$ , indicate the presence of  $sp^2$ -hybridized carbon atoms within the GO structure [28]. Additional peaks, such as the C–O stretch at 1448  $cm^{-1}$  and  $CH_3$  bending at 1774  $cm^{-1}$ , further confirm the presence of functional groups and chemical bonding within the composite material. These observations suggest that the nanocomposites maintain a well-defined crystalline structure, with GO incorporation affecting the vibrational characteristics. This modification in vibrational properties contributes to the enhanced photocatalytic and electronic performance of the nanocomposites. The vibrational modes observed in the

Raman spectra are in good agreement with the crystalline phases identified by XRD. Specifically, the strong Raman bands at 370, 459, and 644  $cm^{-1}$  correspond to the ZnO  $E_2$  and  $A_1(TO)$  modes, which confirm the hexagonal wurtzite structure also revealed by XRD. Furthermore, the presence of the D and G bands in the GO-containing samples (at  $\sim 1352$  and  $\sim 1617$   $cm^{-1}$ , respectively) supports the effective incorporation of exfoliated graphene oxide, which was consistent with the disappearance of the GO peak in XRD patterns due to its exfoliated/disordered state. This trend is also observed in similar studies, such as [29], where the attenuation of the (002) XRD peak and preservation of Raman bands confirmed structural disruption and layer separation in graphite-based systems. Together, these findings reinforce the successful formation of a single-phase  $K_2FeO_4/ZnO/GO$  nanocomposite.

#### 4.7. Dielectric Analysis

The dielectric properties of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites were examined using LCR (inductance, capacitance, resistance) analysis. The dielectric constant ( $\epsilon'$ ) and dielectric loss ( $\tan \delta$ ) were measured across various frequencies and calcination temperatures (773 K, 873 K, and 973 K), as shown in Figures 9 and 10. For all samples, the dielectric constant decreases with increasing frequency, a typical behavior in dielectric materials that indicates interfacial polarization and dipole relaxation mechanisms within the composite. The incorporation of GO into the  $K_2FeO_4/ZnO$  matrix results in a higher dielectric constant at lower frequencies, likely due to increased interfacial polarization introduced by the GO layers. Additionally, dielectric loss decreases with frequency, suggesting reduced energy dissipation at higher frequencies. These trends indicate that the dielectric properties of the nanocomposites can be modulated by adjusting both the calcination temperature and GO content, making these materials promising for use in capacitors, sensors, and other electronic devices. Notably, the  $K_2FeO_4/ZnO/GO$  sample calcined at 973 K exhibited a distinctive dielectric loss profile compared to other samples. This anomaly is attributed to high-temperature-

induced structural rearrangements, possible partial reduction of GO, and enhanced interfacial polarization effects. These factors collectively may lead to increased conductive pathways and dipolar relaxation losses, as supported by similar observations in GO-based nanocomposites reported in the literature [29].

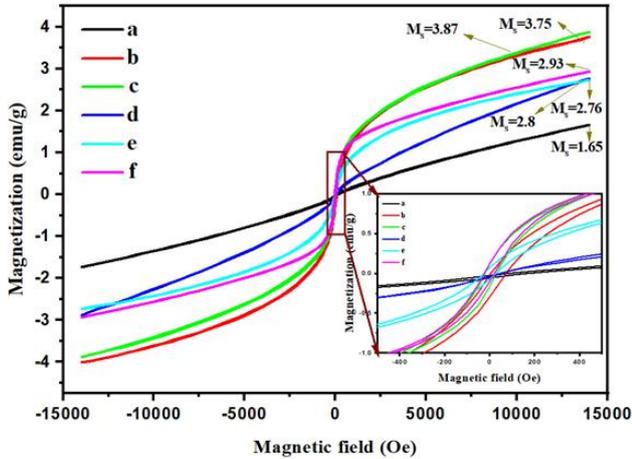


Fig. 7. VSM patterns of calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.

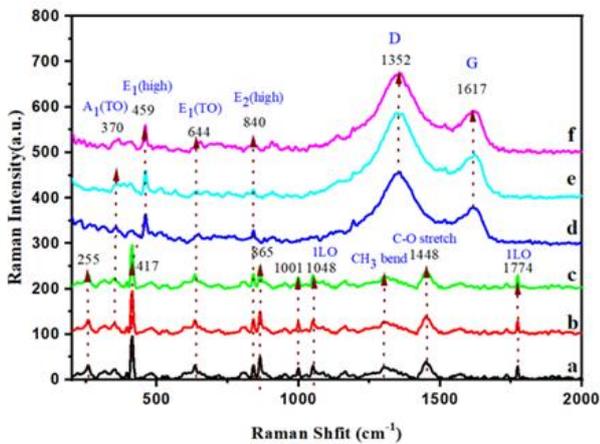


Fig. 8. RAMAN patterns of patterns of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823 and (f) 923 K.

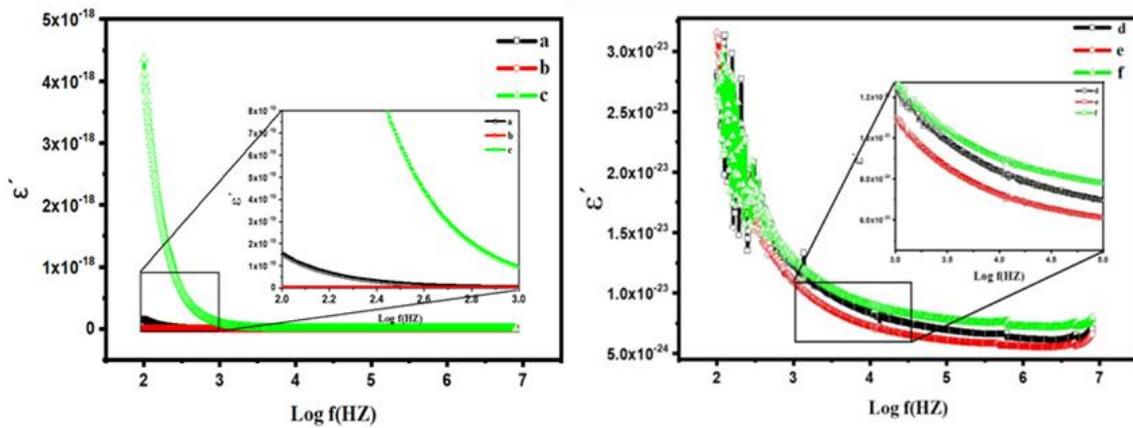
#### 4.8. Photocatalytic Analysis

The photocatalytic properties of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites were evaluated by assessing their ability to degrade Methylene Blue (MB) dye under visible light. Absorption spectra of the nanocomposites, calcined at 723, 823, and 923 K, were recorded in the wavelength range of 200–800 nm over a 90-minute irradiation period, with measurements taken at 15-minute intervals (Fig. 11). A reduction in the MB absorption peak intensity over time indicates effective photocatalytic degradation. Notably, samples calcined at 823 K and 923 K showed a more pronounced reduction in absorption, suggesting enhanced photocatalytic activity at higher calcination temperatures. The increase in photocatalytic properties of nanoparticles at high

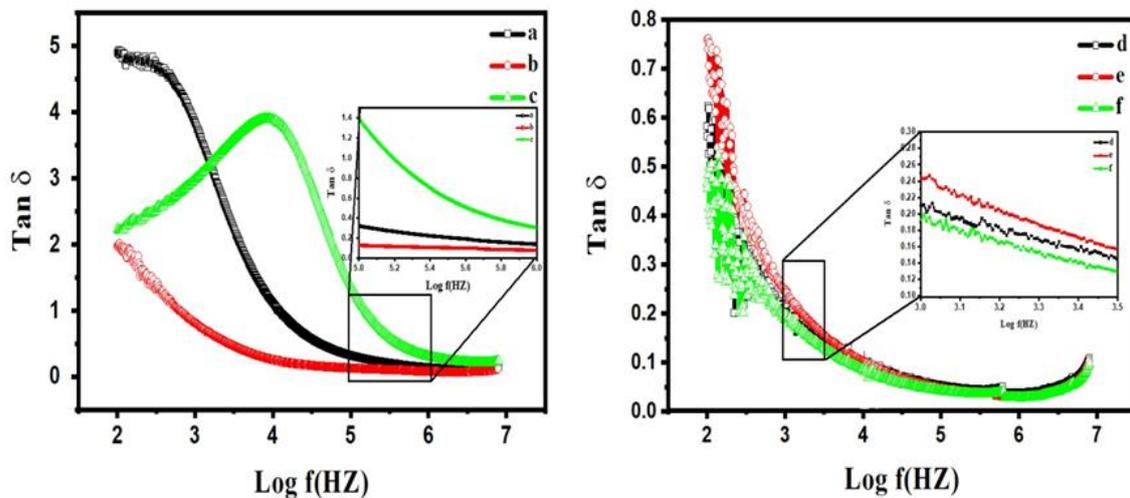
calcination temperatures is generally attributed to several interconnected factors such as improved crystallinity, increased particle size, reduced surface defects and removal of impurities [30] that improve their ability to absorb light and facilitate charge separation and transfer, leading to enhanced generation of reactive oxygen species (ROS). Calcination can increase particle size, thereby influencing the band gap and light absorption properties. In  $K_2FeO_4/ZnO$  nanoparticles, thermal calcination induces merging through grain boundary diffusion driven by zinc or oxygen defects [31]. The  $K_2FeO_4/ZnO/GO$  nanocomposite exhibited the most significant decline in absorbance, highlighting the synergistic effect of GO in enhancing light absorption and charge separation. The photocatalytic degradation efficiency (DE%) was calculated using the formula  $DE(\%) = [(C_0 - C) / C_0] \times 100$ , where  $C_0$  is the initial concentration of MB, and  $C$  is the concentration after irradiation. Figure 12(a) illustrates the DE% of MB as a function of irradiation time for both nanocomposites. Results show that both  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites exhibit photocatalytic activity, with  $K_2FeO_4/ZnO/GO$  achieving significantly higher degradation efficiency than  $K_2FeO_4/ZnO$  alone. After approximately 60 minutes, degradation rates stabilize, indicating that the photocatalytic process is nearing maximum efficiency. The DE% continues to increase with irradiation time, especially for  $K_2FeO_4/ZnO/GO$ , suggesting that GO enhances photocatalytic performance, particularly at higher calcination temperatures. In the context of first-order kinetic reactions, the expression  $\ln(C_0/C)$  is used to describe the concentration changes of MB over time. Figure 12(b) shows that the composite containing GO has a greater increase in  $\ln(C_0/C)$  over time than the composite without GO, confirming more efficient MB degradation in the presence of GO. At a calcination temperature of 923 K, this effect is particularly strong, likely due to improved crystallinity and surface properties of the nanocomposite, which further enhance photocatalytic performance. The combination of GO with  $K_2FeO_4/ZnO$  not only improves degradation efficiency but also accelerates reaction kinetics, underscoring the critical role of GO in optimizing the photocatalytic properties of the composite for environmental remediation applications. In summary, the improved photocatalytic activity of GO-based nanoparticles arises from a combination of enhanced light absorption, efficient charge separation and transport, increased surface area, and synergistic effects with other material [32]. GO acts as an electron acceptor, efficiently separating photogenerated electrons and holes. The electrons and the holes preventing recombination and improving the photocatalytic efficiency [33]. Table 3 provides a detailed breakdown of MB degradation percentages at 180 minutes of irradiation for each calcination temperature (723, 823, and 923 K), indicating that higher calcination temperatures yield greater MB degradation and enhanced photocatalytic performance.

**Table 2.** Magnetic parameters obtained from VSM spectrum.

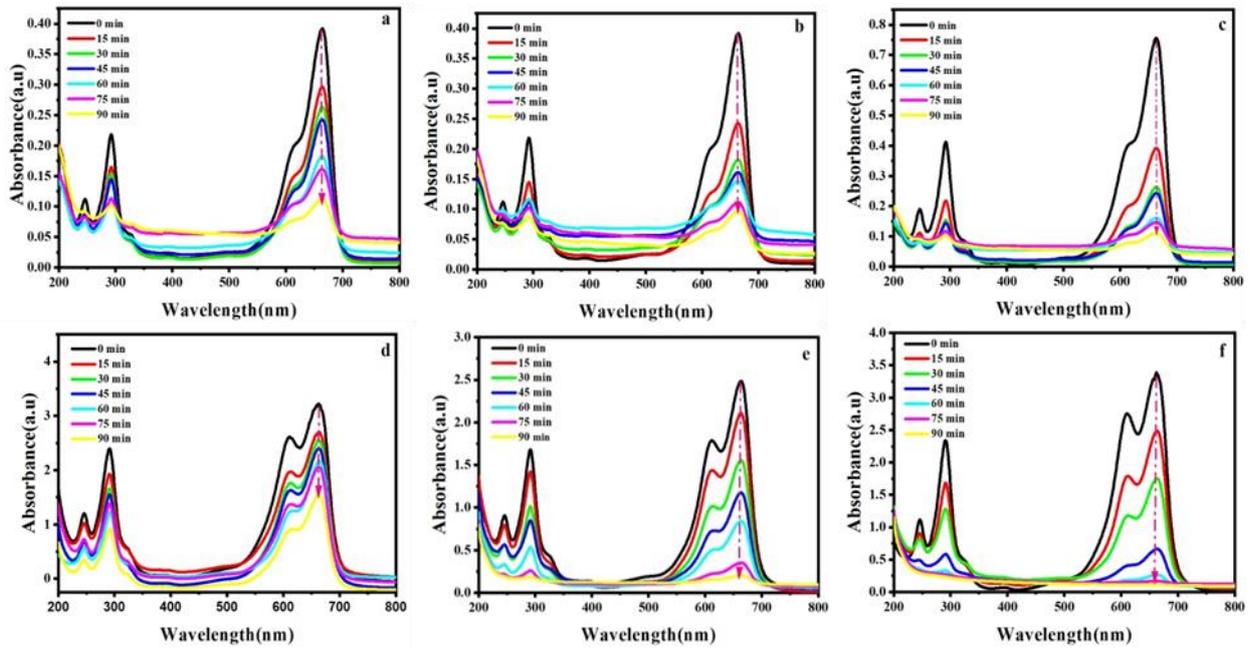
Samples	$M_s$ (emu/g)	$M_r$ (emu/g)	$(M_r/M_s)$	$H_c$ (Oe)
$K_2FeO_4/ZnO$ (T=723 K)	1.65	~0	~0	~74
$K_2FeO_4/ZnO$ (T=823 K)	3.75	~0/05	0/01	~70
$K_2FeO_4ZnO$ (T=923 K)	3.87	~0/13	0/03	~27
$K_2FeO_4/ZnO/GO$ (T=723 K)	2.18	~0/02	0/009	~50
$K_2FeO_4/ZnO/GO$ (T=823 K)	2.76	~0/07	0/02	~25
$K_2FeO_4/ZnO/GO$ (T=923 K)	2.93	~0/12	0/04	~14



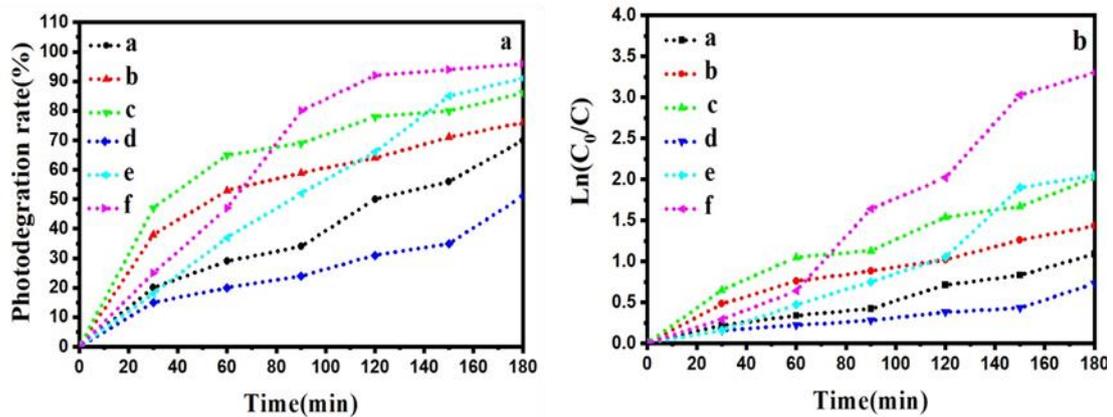
**Fig. 9.** Relationship between: dielectric constant and frequency of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.



**Fig. 10.** Relationship between: dielectric loss tangent and frequency of patterns of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.



**Fig. 11.** Absorption spectra of the MB solution taken at different photocatalytic degradation times using of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.



**Fig. 12.** (A) photocatalytic percent degradation of MB of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823 and (f) 923 K, and (B) Photodegradation rate of MB in solution of  $K_2FeO_4/ZnO$  calcined at (a) 723, (b) 823, and (c) 923 K, and  $K_2FeO_4/ZnO/GO$  calcined at (d) 723, (e) 823, and (f) 923 K.

**Table 3.** The photocatalytic degradation efficiency and the pseudo first order rate constant values for the of  $K_2FeO_4/ZnO$  and  $K_2FeO_4/ZnO/GO$  nanocomposites.

Samples	Calcination Temp. (K)	Irradiation time (min)	Photodegradation rate (%)	$\ln(C_0/C)$
$K_2FeO_4/ZnO$	T=723	180	70	1.09
$K_2FeO_4/ZnO$	T=823	180	75	1.46
$K_2FeO_4/ZnO$	T=923	180	86	2.02
$K_2FeO_4/ZnO/GO$	T=723	180	51	0.73
$K_2FeO_4/ZnO/GO$	T=823	180	91	2.5
$K_2FeO_4/ZnO/GO$	T=923	180	96	3.4

#### 4.9. General Remarks and Future Outlook

This study highlights the multifunctional behavior of  $K_2FeO_4/ZnO/GO$  nanocomposites, where the incorporation of GO markedly enhanced photocatalytic activity under visible light due to improved charge separation, interfacial contact, and light absorption.

The highest degradation efficiency of methylene blue was achieved at 923 K, while soft magnetic behavior at lower temperatures supports potential applications in magneto-photocatalytic systems. However, the limited impact of GO on dielectric permittivity suggests the need for further interfacial engineering or incorporation of additional dielectric-active phases. To expand the

functionality and efficiency of these composites, several future directions are proposed:

**Layered architecture optimization:** Inspired by  $\text{SnO}_2/\text{Ag}/\text{SnO}_2$  multilayers, designing  $\text{K}_2\text{FeO}_4/\text{ZnO}/\text{GO}$  films with graded or multilayered structures may improve visible light transmission, infrared reflectance, and reduce sheet resistance, enabling applications in thermal insulation and optoelectronics [34].

**Biopolymer-based dispersion enhancement:** Glucosamine and chitosan, due to their abundant hydroxyl and amine groups, offer strong coordination with metal oxide surfaces and can prevent nanoparticle aggregation [35]. Glucosamine, a naturally occurring amino sugar, is widely used in pharmaceutical and biomaterial applications due to its biocompatibility and bioactivity [36]. These properties, combined with its functional amine and hydroxyl groups, suggest it could serve as a green modifier or bio-linker in photocatalytic and dielectric nanocomposites. Glucosamine has been shown to control particle dispersion and enhance surface area and photocatalytic activity in  $\text{TiO}_2/\text{graphene}$  composites [37]. Such strategies could be extended to  $\text{K}_2\text{FeO}_4/\text{ZnO}/\text{GO}$  systems for improving catalytic and dielectric behavior.

**Antimicrobial-functional hybrid materials:** Inspired by  $\text{AgFe}_2\text{O}_4/\text{SiO}_2/\text{Passiflora}$ -based nanocomposites, which exhibited superior antimicrobial activity [38],  $\text{K}_2\text{FeO}_4/\text{ZnO}/\text{GO}$  composites could be modified with  $\text{Ag}^+$  ions, ferrite phases, or biocompatible agents to achieve dual photocatalytic-antibacterial functionality, ideal for water purification or biomedical coatings.

**Multifunctional catalytic platforms:**  $\text{Cu}_2\text{V}_2\text{O}_7$  nanoparticles have shown efficient photocatalytic and organic synthetic activity, including recyclability in heterocyclic synthesis [39]. Incorporating such vanadates into the  $\text{K}_2\text{FeO}_4/\text{ZnO}/\text{GO}$  framework may expand its role beyond pollutant degradation to broader green chemical transformations.

**Perovskite-based enhancement:**  $\text{CeFeO}_3/\text{GO}$  nanocomposites exhibited 97% MB degradation in just 25 minutes under visible light, attributed to strong GO- $\text{CeFeO}_3$  interaction and reduced crystallite size [40]. This suggests the potential of Ce- or La-based perovskites as additional components in future designs of  $\text{K}_2\text{FeO}_4/\text{ZnO}/\text{GO}$  nanohybrids for advanced visible-light photocatalysis.

Overall, this research establishes a foundation for tailoring  $\text{K}_2\text{FeO}_4$ -based systems toward high-performance, multifunctional materials applicable in energy conversion, environmental remediation, catalysis, and potentially biomedical technologies.

## 5. Conclusions

This paper presents significant findings regarding the impact of ZnO and GO doping on the structural and electrical properties of potassium ferrate nanostructures. The samples were synthesized using a thermal treatment method at temperatures of 723, 823, and 923 K. The structural characteristics of the synthesized samples were analyzed using various techniques, including X-ray diffraction (XRD), field emission scanning electron microscopy (FESEM), energy-dispersive X-ray

spectroscopy (EDS), transmission electron microscopy (TEM), ultraviolet-visible (UV-Vis) spectroscopy, vibrating sample magnetometry (VSM), Raman spectroscopy, and Fourier transform infrared (FTIR) spectroscopy. XRD and TEM analyses confirmed the formation of all samples with minimal impurities, and the average crystallite size was observed to range between 30 nm and 40 nm. The dielectric constant and dielectric loss were measured at room temperature across a frequency range of 4 Hz to 8 MHz using an LCR meter. Our findings reveal that the addition of graphene oxide (GO) did not result in a significant enhancement in dielectric permittivity compared to potassium ferrate/ $\text{ZnO}$  nanocomposites. Therefore, potassium ferrate-based nanocomposites, given their favorable dielectric properties, are promising candidates for a wide range of applications, including devices operating at microwave frequencies, various optical and microelectronic applications, and as materials for microwave absorption. The synthesized  $\text{K}_2\text{FeO}_4/\text{ZnO}$  nanocomposites with GO sheets demonstrated superior photocatalytic activity compared to pure  $\text{K}_2\text{FeO}_4/\text{ZnO}$  nanoparticles, as evidenced by their more efficient degradation of MB dye under visible light irradiation.

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## Conflicts of Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## Authors Contribution Statement

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**Manuscript writing and editing:** Azam Hashemi, Mahmoud Naseri

**Supervision and project administration:** Mahmoud Naseri

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